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## INVESTIGATION OF THE FIRST-ORDER MAGNETIC TRANSFORMATION IN Mn<sub>3</sub>Pt

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BUDAPEST

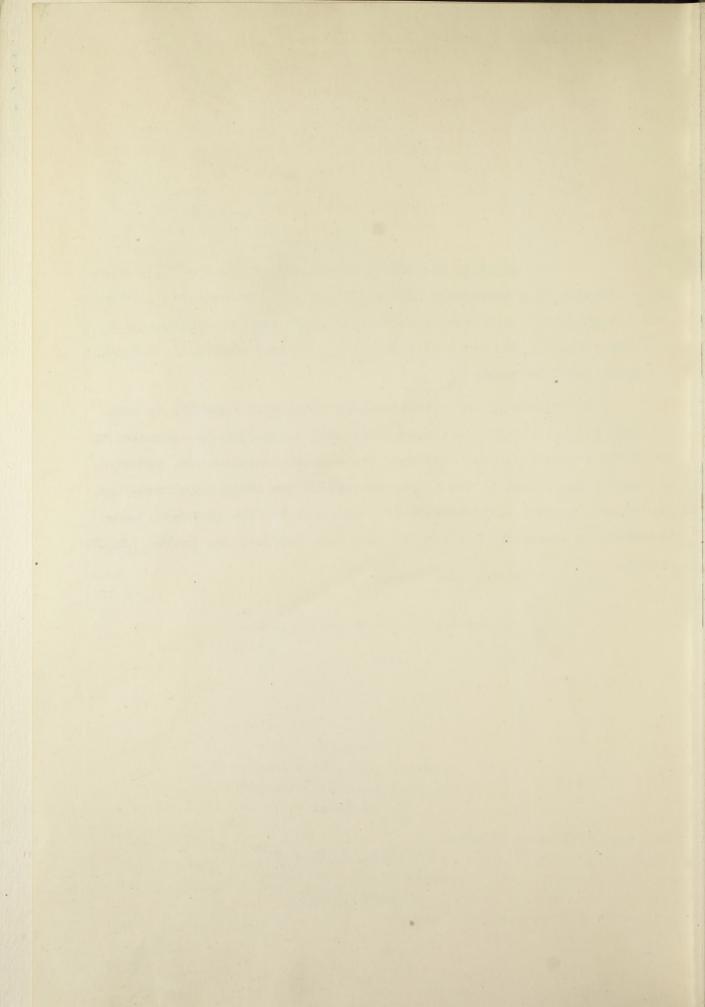
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Eng.szám: 883-5677/966 Megjelent: 120 példányban Felelős: Gyenes Imre Abstract The effect of composition on the first-order transformation from triangular into collinear antiferromagnetic structure /AF-AF transformation/ occuring in ordered Mn<sub>3</sub>Pt has been investigated on ternary Mn<sub>3</sub>Pt<sub>1-x</sub>Rh<sub>x</sub> and binary Mn<sub>3+y</sub>Pt<sub>1-y</sub> alloys using X-ray and neutron diffraction methods.

On increasing the concentrations x and y an increase in both the Néel and the transition temperatures with a simultan eus decrease in the lattice parameter were observed. The magnetic moment of Mn remained the same irrespective of the concentration and the AF-AF transformation. The results suggest the existence of a critical lattice parameter value at which the transformation occurs, like that observed for the Mn<sub>2-x</sub>Cr<sub>x</sub>Sb system.



Recently we have observed a first-order magnetic transformation in Cu<sub>3</sub>Au type ordered Mn<sub>3</sub>Pt. The magnetic structure transformed from triangular to collinear antiferromagnetic with incerasing temperature. The atomic and magnetic unit cells in the triangular structure are identical, therefore the magnetic neutron scattering appears as a contribution to the nuclear reflections. In the collinear structure the magnetic unit cell is doubled, the neutron diffraction pattern is characterized by magnetic superreflections indexed by half-integers. The transformation can be easily followed up by measuring the temperature dependence of the two types of magnetic reflections. The transition is accompanied by an abrupt change in the lattice parameter as well as by thermal hysteresis. Mn<sub>3</sub>Rh of the same type, however, was found to have triangular structure up to the Néel temperature 1,2.

Continuing these investigations the effect of composition on the AF-AF transformation was studied on alloys in the ternary /pseudo-binary/
Mn<sub>3</sub>Pt<sub>1-x</sub>Rh<sub>x</sub> and the binary Mn<sub>3+y</sub>Pt<sub>1-y</sub> systems by X-ray and neutron diffraction methods.

Alloys of x=0; 0,05; 0,1; 0,2; 0,5; 1,0 and y=0,09; -0,07 concentrations were prepared from 99,9% purity metals by melting. The loss in weight was about 0,2% in each case. The Mn concentration was checked by chemical analysis. The ingots were filed, then annealed at .700°C for 120 hours in evacuated quartz tube. X-ray diffraction photographs taken at room temperature showed a highly ordered Cu<sub>3</sub>Au type single phase with values of lattice parameter a given in Table I.

At  $77^{\circ}$ K the neutron diffraction patterns of each specimen indicate triangular structure with magnetic moments of Mn listed in Table I. The magnetic moment in the collinear phase could be evaluated only for y=0 and -0,07 as 2,4±0,3 and 2,8±0,3  $\mu_{\rm B}$  at 380 and 340°K respectively. The transition temperature  $T_{\rm t}$  of the AF-AF transformation and the Néel temperature  $T_{\rm t}$  were determined from the temperature dependence of the (110) and( $10\frac{1}{2}$ ) reflections associated with triangular and collinear structures, respectively. The results for (110) are shown in Fig. 1. If observable, ( $10\frac{1}{2}$ ) changes oppositely. The values of  $T_{\rm t}$  and  $T_{\rm N}$  are included in Table I. In spite of the abrupt change and thermal hysteresis in /110/,  $T_{\rm N}$  could not be determined in the y=0,09 specimen since ( $10\frac{1}{2}$ ) could not be observed. This is probably due to the closeness of the values of  $T_{\rm t}$  and  $T_{\rm N}$ . The magnetic phase diagram of the Mn<sub>2</sub>Pt<sub>1-X</sub>Rh<sub>X</sub> system is given in Fig. 2. As seen, the collinear structure is stable only in a limited concentration and temperature range. It would be more stable in the binary

alloys at lower Mn concentrations if the ordered crystal structure existed  $^3$ . The variations of  $\underline{a}$  with temperature was measured by high temperature X-ray diffractometer on specimens of y=0 and -0,07. The curves in Fig.3 show a discontinuity at  $T_t$  as well as a break at  $T_N$ . An anomalous behaviour between  $T_t$  and  $T_N$  is apparent for y = 0,07.

On increasing the x and y concentrations an increase in both  $T_t$  and  $T_N$  with a decrease in  $\underline{a}$  can be observed. Within the experimental accuracy the magnetic moment of Mn does not change with concentration nor during the AF-AF transformation. The simultaneous variation of  $T_t$  and  $\underline{a}$  suggest the existence of a critical lattice parameter value at which the transition occurs, similarly to the  $Mn_{2-x}Cr_xSb$  system the relation between  $T_t$  and  $\underline{a}$  is not linear,  $T_t$  is more sensitive to changes in  $\underline{a}$  if close to  $T_N$ , this may be attributed to the anomaly in  $\underline{a}$  in Fig. 3. The different values of  $T_t$  in x = 0.2 and y = 0.09, for which  $\underline{a}$  is the same, could be explained by different thermal expansion coefficients in the ternary and binary alloys. In order to clarify the role of the lattice parameter in the AF-AF transformation, X-ray measurements on further specimens are in progress.

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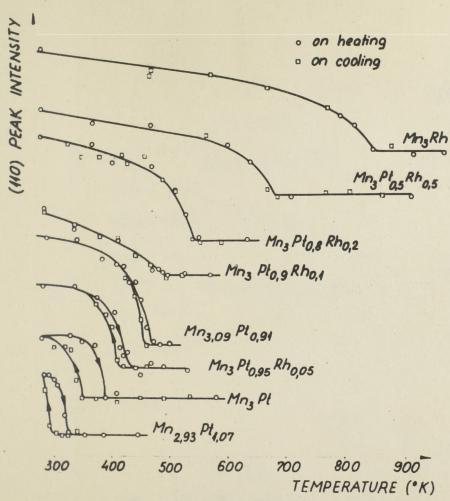


Fig.1. Temperature dependence of the (110) reflection

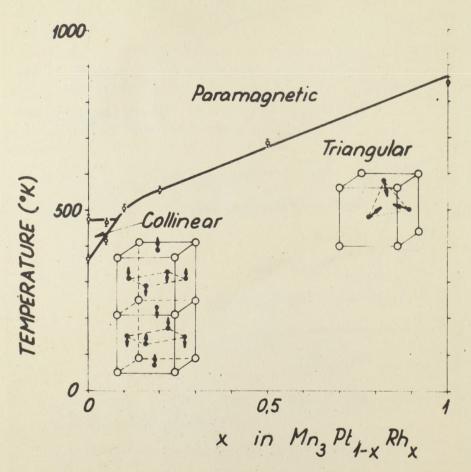


Fig.2. Magnetic phase diagram of the Mn\_3Pt\_1-xRh\_x system.

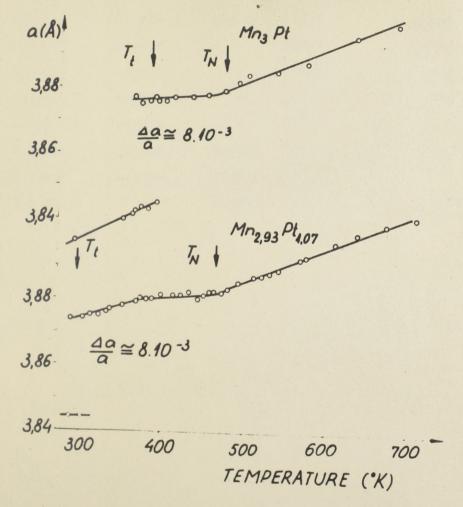


Fig. 3. Temperature dependence of the lattice parameter for y=0 and -0,07

Table I Values of lattice parameter, transition temperatures and magnetic moment

Specimen	a/R/	T <sub>t</sub> /°K/	T <sub>N</sub> /°K/	M/ MB/
Mn <sub>3</sub> Pt	3,833	365 <u>+</u> 10	475 <u>+</u> 10	3,0 <u>+</u> 0,3
Mn <sub>3</sub> Pt <sub>0,95</sub> Rh <sub>0,05</sub>	3,833	415 <u>+</u> 10	465 <u>+</u> 10	3,5 <u>+</u> 0,4
Mn <sub>3</sub> Pt <sub>0,9</sub> Rh <sub>0,1</sub>	3,832	-	505 <u>+</u> 10	2,8 <u>+</u> 0,3
Mn3Pt0,8Rh0,2	3,828	-	555 <u>+</u> 10	3,2 <u>+</u> 0,4
Mn <sub>3</sub> Pt <sub>0,5</sub> Rh <sub>0,5</sub>	3,820	-	685 <u>+</u> 10	3,5 <u>+</u> 0,4
Mn <sub>3</sub> Rh	3,813	_	855 <u>+</u> 10	3,6 <u>+</u> 0,4
Mn <sub>2</sub> ,93 <sup>Pt</sup> 1,07	3,844	315 <u>+</u> 10	460 <u>+</u> 10	3,4+0,4
Mn <sub>3</sub> Pt	3,833	365 <u>+</u> 10	475 <u>+</u> 10	3,0 <u>+</u> 0,3
Mn <sub>3</sub> ,09 <sup>Pt</sup> 0,91	3,827	455 <u>+</u> 10		3,2 <u>+</u> 0,4

